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# FTIR traced modification of defect-engineered UiO-66 for enhanced accessibility of zirconium sites

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#### ABSTRACT

This research focuses on identifying the accessibility of active sites within the defect-engineered UiO-66 framework. The task is particularly challenging due to reversible changes in the framework during dehydroxylation: the loss of  $\mu_3$ -OH groups with simultaneous reduction of the Zr coordination number and the possible creation of Zr<sup>4+</sup> Lewis acid sites in defect MOFs. We used in-situ FTIR and XANES analyses, as well as interaction with probe molecules, to monitor the changes in Zr coordination and the host-guest interaction. The defects were introduced using benzoic acid as a modulator, which coordinated to Zr<sup>4+</sup> in defective pores. Our results showed that the UiO-66 sample synthesized with benzoic acid contained defects, but these were concealed under benzoate residues and thus inaccessible. Standard washing and heating did not remove benzoate anions. Dehydroxylation of the sample leads to the development of "hidden" Lewis acidity: some Zr<sup>4+</sup> sites were not able to form complexes with the weak base CO, but they interact with the stronger bases acetonitrile. Additionally, insitu XANES analysis revealed that the effect of acetonitrile adsorption is similar to that of water rehydration. Treatment of the sample with HCl and DMF led to the replacement of benzoates with formate ions, exposing the bare  $Zr^{4+}$  sites within the defective pores. These cationic sites acted as true Lewis acids and were able to coordinate both CO and acetonitrile. Our findings emphasize that the active sites in UiO-66 highly depend on synthesis conditions and post-synthetic treatments. Comprehensive site-specific methods are crucial for accurately predicting and identifying these active sites.

# 1. Introduction

Zirconium-based metal-organic frameworks (MOFs) represent a class of porous materials known for their exceptional thermal and chemical stability. These MOFs consist of Zr-O clusters coordinated with carboxylic linkers, forming strong covalent bonds and achieving high coordination numbers of  $Zr^{4+}$ , contributing to their remarkable stability. UiO-66, a prominent family member, was first reported in 2008 [1]. It is composed of  $Zr_6O_4(OH)_4$  clusters linked by 12 terephthalate linkers (BDC), resulting in a cubic structure containing both tetrahedral and octahedral pores [2,3]. UiO-66 has paved the way for a diverse family of isoreticular MOFs with various linkers, typically dicarboxylate acids featuring carboxylic groups in the para-position [2,4,5].

The UiO-66 comprises  $Zr_6O_4(OH)_4$  clusters (Fig. 1b). However, treatment under vacuum at elevated temperatures, referred to as activation, leads to removing two water molecules and transforming the

cluster to  ${\rm Zr}_6O_6$  (Fig. 1a). After activation, the cluster no longer contains  $\mu_3$ -OH groups, and zirconium ions reduce their coordination numbers from 8 to 7 [6]. Notably, this process is reversible, and the samples can be rehydrated.

In the ideal structure of UiO-66, zirconium-oxygen clusters coordinate with 12 BDC linker molecules. This high connectivity renders these MOFs exceptionally tolerant to defects. A coordination modulation approach was proposed to intentionally introduce defects [7–11]. In this approach, a monocarboxylic acid serves as a modulator and is added to the reaction mixture alongside the linker, with a higher modulator concentration than the linker. The modulator and linker compete to bond with  $Zr^{4+}$  ions within the reaction mixture. The higher modulator concentration favors its interactions with the inorganic node. However, the modulators possess only one carboxylic group and cannot bridge two  $Zr_6O_6$  clusters. Consequently, crystal growth occurs when linkers replace modulators. In some cases,  $Zr^{4+}$  ions within the MOF crystal

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retain modulators in their coordination sphere instead of linkers, forming defects. As a result, when a linker is missing, defect pores form, and vacancies appear on two opposite  $Zr_6O_6$  clusters. The effect of modulators on MOF formation varies based on their concentration and properties. It has been reported that increasing the acid strength of the modulator promotes defect formation in MOFs [7,12,13]. Additionally, the size of the modulator molecule plays a role in this process. Benzoic acid, for example, has a molecular size similar to the BDC linker, which leads to the creation of a relatively high number of defects per node. However, at high concentrations of benzoic acid, it becomes difficult for BDC to replace it due to its larger size.

Temporary anions balance the excess positive charge of  $Zr^{4+}$  ions in these defect sites. When UiO-66 is synthesized in DMF, formate ions may be present, as DMF hydrolyzes in the presence of water at elevated temperatures, producing dimethylamine and formic acid. The formate group can bridge two Zr atoms at a vacancy site [14]. Studies have shown that formate ligands decompose at 473 K under high vacuum. However, other researchers have observed that alcohol treatment of activated UiO-66 forms alkoxides on the nodes [12,14]. Defect sites can be used for post-synthetic modifications when temporary residuals are replaced with desired species [15,16]. However, access to these sites may be blocked by residual modulators. For instance, bulky benzoic acid residues can completely obstruct zirconium sites, making them inaccessible to gas molecules or for post-synthetic modifications [17,18].

In summary, UiO-66 is a complex system containing various potential active sites, which can be altered during the activation procedure or blocked by modulator anions. Generally, the active sites within Zr-O clusters of UiO-66 can be considered in three categories. Firstly, Zr<sub>6</sub>O<sub>4</sub>(OH)<sub>4</sub> clusters initially contain  $\mu_3$ -OH groups with weak Brønsted acidity [17,19]. The activation process removes these sites, reducing Zr's coordination number and allowing other molecules to replace the segregated  $\mu_3$ -OH groups and form bonds with Zr<sup>4+</sup>. Lastly, Zr<sup>4+</sup> ions in defect pores, initially bonded with modulator molecules, become unsaturated and acquire Lewis acidic properties if these molecules are removed.

In this research, we assessed the accessibility of active sites in UiO-66 using FTIR analysis with probe molecules as the primary tool, complemented by XANES spectroscopy and other basic characterization methods. We compared two samples: the first was UiO-66 synthesized with benzoic acid as a modulator. This sample contained defective pores with benzoate ions bonded to zirconium sites. The second sample was obtained by treating the first sample with a DMF and HCl mixture to remove benzoate residues from the defective pores. Both samples were investigated in two states: hydrated and dehydrated. This comparison was used to evaluate and distinguish the effects of hydrationdehydration and compensating ions in defective pores on the composition and accessibility of active sites in the MOF.

# 2. Materials and methods

The starting materials zirconium tetrachloride (ZrCl<sub>4</sub>), 1,4-benzene dicarboxylic acid (H<sub>2</sub>BDC), N, N-dimethylformamide (DMF), benzoic acid (BA), hydrochloric acid (37 %) and methanol were purchased from Alfa Aesar and used without additional purification. Distilled water was purified via a Simplicity UV ultrapure water system.

# 2.1. UiO-66-BA

The samples were synthesized using the original procedure [15]. Briefly,  $ZrCl_4$  was dissolved in DMF, and deionized water was added to the solution. BA was mixed in and stirred at room temperature until a clear solution was achieved.  $H_2BDC$  was added and dissolved. The molar ratio  $ZrCl_4$ :  $H_2BDC$ :  $H_2O$ : BA: DMF was 1: 1 : 3: 10 : 300. The reaction mixture was then placed in an oven and heated at 120 °C for 24 h. After synthesis, a white precipitate was collected by centrifugation and washed twice with pure DMF and methanol. Finally, the washed precipitate was dried at 60 °C overnight to obtain the UiO-66-BA sample.

# 2.2. UiO-66-FA

According to our previous research, using benzoic acid as a modulator produces defective pores, but its residues block Zr sites. To overcome this limitation, we applied a modification procedure based on [18] for the post-synthetic treatment of the UiO-66-BA sample. In this process, 50 mg of UiO-66-BA sample was combined with 15 mL of DMF and 0.625 mL of 8M HCl. The mixture was stirred thoroughly and loosely covered with a watch glass. Subsequently, the flask and its contents were heated at 100 °C for 24 h in a preheated oven. After the treatment, a microcrystalline powder formed, which was separated from the mixture via centrifugation. The powder was then washed three times with fresh



**Fig. 1.** Schematic representation of the zirconium-oxygen cluster of UiO-66. Part (a) represents the dehydrated  $Zr_6O_6$  cluster, while part (b) represents the hydrated  $Zr_6O_4(OH)_4$  cluster. Zirconium ions are depicted as green spheres, oxygen atoms in the clusters are shown as pink spheres, and bonds inside the clusters are highlighted with blue solid lines. Positions of linker molecules are depicted as carboxylic groups: gray spheres represent carbon atoms, dusty-pink spheres represent oxygen, and dashed gray lines represent bonds in carboxylic groups. Both clusters are presented with defective sites. One linker molecule is missing in each of them, and zirconium is coordinated with temporary compensating ions, shown as purple spheres. The crystal structure was elaborated according to crystallographic data from Ref. [11].

DMF. Lastly, the precipitate was separated from the solvent by centrifugation and dried overnight in an oven set to 60  $^{\circ}$ C. Following this treatment, the sample was referred to as UiO-66-FA in the text.

## 2.3. Characterization

X-ray powder diffraction (XRD) profiles were obtained using a D2 PHASER diffractometer (Bruker Corporation, Germany) in the 20 range of 5–90° with a step size of 0.01 and CuK $\alpha$  radiation with a wavelength ( $\lambda$ ) of 1.5417 Å. The profile analysis was conducted using Jana 2006 software [20].

For ATR-IR spectra measurements, an ex-situ analysis was performed on a Bruker Vertex 70 spectrometer, covering a range from 5000 to 500  $\rm cm^{-1}$  with a resolution of 1 cm<sup>-1</sup>. The data was accumulated through 64 scans using an MCT detector and a Bruker Platinum ATR attachment.

 $\rm N_2$  sorption isotherms were recorded at -196 °C applying ASAP2020. Prior to the measurements, samples were degassed for 10 h at 150 °C in a dynamic vacuum. Specific surface areas (SSA) were calculated according to the BET model. Pore size distribution was calculated using the adsorption branch of the respective isotherm according to the Horvath-Kawazoe model for cylindrical pores.

In situ FTIR spectra of the self-supporting pellets were recorded using a Nicolet 6700 FTIR spectrometer. For these measurements, 64 scans were accumulated at a spectral resolution of 2 cm<sup>-1</sup>. The self-supporting pellets were prepared from the sample powders and analyzed directly in a specially designed IR cell, enabling measurements at both ambient and low (approximately -173 °C) temperatures. The IR cell was connected to a vacuum adsorption apparatus with a residual pressure below  $10^{-3}$ Pa. Before the adsorption experiments, the samples were activated by evacuation at different temperatures. The adsorption experiments were conducted by introducing either 5 mbar of CO (Merck, purity 99.5 %) or CD<sub>3</sub>CN (Merck, deuteration degree 99.96 %) directly into the IR cell. Afterward, the adsorbed gases were diluted, and the cell was evacuated for different durations as part of the experimental procedure.

The NMR spectra were recorded on a Bruker AVANCE NEO 600 MHz spectrometer equipped with a 5 mm Prodigy probehead at a proton operating frequency of 600.18 MHz. The spectra were measured without spinning the sample at a temperature of 298 K. The NMR spectra for quantitative analysis were measured with a pulse sequence with water suppression zgesgppe using the following experimental parameters, optimized for quantitative measurements: 90° hard excitation pulse, spectral width 10.96 ppm, 64 K time domain data points, 16 scans, 10s water suppression time and a relaxation delay of 50 s to ensure complete relaxation for all signals. The spectra were Fourier transformed after zero filling to 128 K, giving a digital resolution in the frequency domain of 0.10 Hz/pt. Careful manual phase and baseline corrections were made prior to signal integration. Before conducting measurements, UiO-66 samples were dissolved in a deuterated alkaline medium (1 M NaOD in  $D_2O$ ) following the method described by Shearer et al. [13]. The powder was added to the alkaline solution, thoroughly mixed, and left for 24 h. Afterward, the suspension was centrifuged, separating the solid and liquid components. All organic compounds were transferred into the liquid phase as their corresponding sodium salts and used for NMR analysis. The white precipitate contained hydrated ZrO<sub>2</sub>.

The Zr K-edge XANES spectra were measured at the Structural Materials Science end-station [21] at the Kurchatov Synchrotron Radiation Source (Moscow, Russia). The Si(220) channel-cut crystal monochromator was selected to obtain  $\Delta E/E = 2 \cdot 10^{-4}$  energy resolution. Signals were detected using ionization chambers. Pellets of the samples mixed with cellulose as an agent for binding and dilution have been prepared to give transmission signals above the Zr absorption edge of µd of about 1. The samples were transferred to a custom in situ cell equipped with a gas line and a heater. Data were averaged over three scans, measured at the same position of a pellet for each sample. The standard data reduction was performed using the Demetra package [22]. The energy was calibrated to the first infection point of a metal Zr spectrum to 18.998 keV (the first maximum of the first derivative). Data were pre- and post-edge background corrected with a linear and a polynomial function, respectively, and were normalized to an edge jump of one to enable comparison of spectra using the Athena program of the Demetra software package. The experiment was performed as follows: i) The measuring cell was evacuated to a pressure of  $5.5 \times 10^{-5}$  bar and heated to  $250 \,^{\circ}$ C to activate the UiO-66 structure. ii) The cell was then cooled to room temperature, and water vapor was introduced to ensure complete cleaning of the solvents. iii) The cell was evacuated again to a pressure of  $5.5 \times 10^{-5}$  bar. iv) Acetonitrile vapor was introduced into the cell to occupy the free positions of the active zirconium centers. v) The cell was once again evacuated and heated to  $250 \,^{\circ}$ C to create active centers and remove acetonitrile. vi) Finally, the sample was cooled to room temperature at a pressure of  $5.5 \times 10^{-5}$  bar, and acetonitrile was reintroduced into the cell.

Raman scattering experiments were conducted using an inVia Raman microscope. Measurements utilized the 532 nm (green) diode laser. Spectral analysis was carried out using the Origin 9 software. Spectra were collected in the  $550-1700 \text{ cm}^{-1}$  range.

## 3. Results

# 3.1. Crystal structure and thermal analysis

Fig. 2a displays the XRD patterns of UiO-66-BA and UiO-66-FA samples. Both samples showed a single-phase UiO-66-type structure. All reflections were identified in a cubic symmetry, with the space group Fm-3m. Despite the post-synthetic treatment, there was no structure collapse or formation of any new phases (Table S1, Fig. S1 in SI). The UiO-66-BA sample exhibited a slightly larger lattice constant compared to the UiO-66-FA sample (Table 1). This effect is believed to be linked to the removal of guest molecules and modulator residuals from the pores during the treatment.

Fig. 2b demonstrates the TGA and DSC curves of UiO-66-BA and UiO-66-FA samples. Both of them contained three steps of weight loss. The first step occurs in the temperature range of 50–150 °C. It corresponds to the removal of physically adsorbed water molecules. The next step in the temperature range of 150–350  $^\circ \rm C$  refers to the elimination of DMF and other bonded guest molecules. A slight inflection at a temperature of about 250 °C is associated with the dehydroxylation of the samples and transformation of the  $Zr_6O_4(OH)_4$  cluster to the form of  $Zr_6O_6$ . The theoretical weight loss of ideal UiO-66 is 55.6 % and 54.6 % according to the formula unit  $Zr_6O_4(OH)_4(BDC)_6$  and  $Zr_6O_6(BDC)_6$  respectively (see calculation details in SI, Part 2). The UiO-66-BA sample demonstrated a weight loss of 55.8 %, which is very close to the theoretical value. The UiO-66-FA sample exhibited lower weight loss, estimated at 42.9 %. It could indicate that a more bulky BA modulator was replaced with a less bulky one. These results are in good agreement with a previous report [18].

# 3.2. Porosity

Low-temperature N<sub>2</sub> sorption isotherms are shown in Fig. 3a. Both isotherms correspond to type I according to the IUPAC notification. These shapes of isotherms are typical of microporous materials. In the P/P<sub>0</sub> range of 0.8–1.0, we noticed hysteresis loops indicating capillary condensation of N<sub>2</sub> in spaces between monodisperse nanoparticles. Specific surface areas were calculated according to the BET model and were estimated as  $1335 \text{ m}^2 \text{ g}^{-1}$  for the UiO-66-BA sample and 1715 m<sup>2</sup> g<sup>-1</sup> for the UiO-66-FA sample (For detailed information, refer to the SI, part 3). So, modification of the UiO-66-BA sample resulted in an increase of its specific surface area by 1.3 times. Fig. 3b represents pore size distribution for pristine and modified UiO-66 samples. The first two picks at 8 and 10 Å correspond to regular pores in the UiO-66-FA sample showed an additional pick at about 20–25 Å, indicating



**Fig. 2.** (a) XRD patterns of the UiO-66-BA and UiO-66-FA samples. Profiles were shifted along the intensity axis. Intensities were multiplied by ten for 20-region after dotted lines for better representation. Numbers designate Miller indices. (b) TGA and DSC curves of UiO-66-BA and UiO-66-FA samples. The TGA curves were normalized based on the assumption that each mole of the UiO-66 formula unit converts to 6 mol of  $ZrO_2$  during the calcination process. Additionally, 6 mol of  $ZrO_2$  was considered to be 100 %. For detailed information, refer to the SI. Gray dashed lines show the theoretical positions of the plateau according to formula unit  $Zr_6O_4(OH)_4(BDC)_6$  and  $Zr_6O_6(BDC)_6$ .

#### Table 1

Some characteristics of UiO-66-BA and UiO-66-FA samples. SSA stands for specific surface area. Pore volume was measured at  $P/P_0 = 0.97$ .

| Sample    | Structure   |                   | N <sub>2</sub> sorption |                                  |  |
|-----------|-------------|-------------------|-------------------------|----------------------------------|--|
|           | a, Å        | V, Å <sup>3</sup> | SSA, $m^2 \cdot g^{-1}$ | Pore volume, $cm^3 \cdot g^{-1}$ |  |
| UiO-66-BA | 20.7731 (4) | 8964.1 (3)        | 1345                    | 0.646                            |  |
| UiO-66-FA | 20.7610 (5) | 8948.4 (4)        | 1812                    | 0.796                            |  |

accessible defect pores.

#### 3.3. IR and Raman

Fig. 4 presents the IR and Raman spectra of UiO-66-BA and UiO-66-FA samples. Treatment with HCl and DMF did not cause any significant changes in the fingerprint region of the spectra, suggesting no alterations in the UiO-66 structure (Fig. S4). However, certain bands at 937, 1025, 1068, 1139, and 1177 cm<sup>-1</sup> vanished after treatment. These bands correspond to vibrations of benzoate ions [7,17,23] (see detailed description in Table S2). The spectrum of the UiO-66-FA sample did not contain these benzoate modulator peaks, indicating the removal of the benzoate modulator. A weak band at 1060 cm<sup>-1</sup> in the spectrum of the UiO-66-FA sample can be attributed to the out-of-plane vibrations of C-H bonds in formate ions, which have replaced benzoate ions [24,25].

These findings are also supported by the Raman spectra (Fig. 4b). The bands at 618 and 1005 cm<sup>-1</sup> in the spectrum of the UiO-66-BA sample were assigned to benzoate residuals [26]. These bands were not observed in the spectrum of the UiO-66-FA sample. A shoulder at 1070 cm<sup>-1</sup> in the Raman spectrum of UiO-66-FA could be due to the  $\delta_{as}$  (CH) vibrations of formate ions [26].

# 3.4. In situ FTIR

## 3.4.1. Activation

We traced the activation procedure for UiO-66-BA and UiO-66-FA samples using *in-situ* FTIR spectroscopy. After synthesis, each sample was evacuated at the respective temperature, cooled down under a dynamic vacuum, and placed into an IR beam without exposure to air. Treating the samples in a dynamic vacuum at room temperature removed physically adsorbed water molecules from the pores, revealing a region in the IR spectra from 2600 to 3800 cm<sup>-1</sup>. Fig. 5 shows selected areas for each sample evacuated at room temperature, at 250 °C, and after subsequent rehydration. The rehydration process included two steps: exposure to 5 mbar of water vapor at room temperature, followed by evacuation at room temperature and holding under dynamic vacuum for 15 min. The spectra of both UiO-66-BA and UiO-66-FA samples contained a band at 3674 cm<sup>-1</sup>, attributed to the stretching O-H modes of  $\mu_3$ -OH groups [17,19,26]. Evacuation at 150 °C increased the peak intensity because of destruction of some OH…OH<sub>2</sub> and OH… DMF adducts.



**Fig. 3.** (a) N<sub>2</sub> sorption isotherms of UiO-66-BA (green square markers) and UiO-66-FA (pink circle markers) samples. Filled markers represent adsorption branches of isotherms; empty markers show description ones. (b) Pore size distribution for UiO-66-BA (green plot) and UiO-66-FA (pink plot).



Fig. 4. IR spectra (a) and Raman spectra (b) of samples UiO-66-BA and UiO-66-FA. Dashed lines highlight bands assigned to benzoate ions.

Further increasing the evacuation temperature led to a reduction in peak intensity due to the dehydroxylation process. Consequently, activation at 250 °C resulted in the transformation of the  $Zr_6O_4(OH)_4$  cluster to  $Zr_6O_6$  in both samples (Fig. S6). Rehydration of the samples led to the restoration of the corresponding peak intensity, indicating the reversibility of the dehydration process (Fig. S7a).

It is also noteworthy that after the first rehydration, the position of the  $\mu_3$ -OH peak was slightly shifted, while subsequent dehydroxylation-rehydroxylation procedures did not affect the peak position (Fig. S7a). It has been reported that the position of the  $\mu_3$ -OH peak is strongly influenced by the functional groups of linkers and the compensating anions at defective Zr sites, such as benzoate residues [12]. We infer that this shift is connected to the evacuation of DMF guest molecules from the pores of the MOF. This is confirmed by the fact that some peaks (highlighted with arrows in Fig. 4) disappeared after the first activation and did not reappear during the rehydration process. These peaks were assigned to DMF vibrations (see Table S2).

After the first evacuation at room temperature, we observed peaks in the 3030–3080 cm<sup>-1</sup> region in the spectrum of the UiO-66-BA sample (Fig. 5a). These peaks are assigned to benzoate vibrations (Table S2). Subsequent activation and rehydration procedures did not lead to the loss of these peaks, suggesting that benzoate residues cannot be removed through heating and evacuation. The spectrum of the UiO-66-FA sample after room temperature evacuation did not contain these bands, indicating that the modification procedure successfully removed benzoate residues. The new bands in the 2850–2955 cm<sup>-1</sup> region of the spectrum of the UiO-66-FA sample were attributed to formate ions, which replaced the benzoate ions [18,27]. The removal of terminal ions in Zr-MOFs has been reported to trigger the formation of reactive OH/H<sub>2</sub>O pairs, with characteristic bands located around 3720–3780 cm<sup>-1</sup>, as observed in NU-1000 [28,29]. However, when the UiO-66-FA sample was activated at 250  $^{\circ}$ C, no such bands appeared in the spectra. This does not definitively prove that no formate ions were removed during the evacuation at 250  $^{\circ}$ C; in fact, we suggest that some were. Nevertheless, the sites were not terminated by the reactive OH/H<sub>2</sub>O groups reported in other cases [28,29].

We used NMR spectroscopy to study the obtained samples. First, the samples were dissolved in NaOD in  $D_2O$ , and the solid product was separated, leaving a solution of organic components for NMR analysis [13]. The spectra were normalized, with the BDC peak set at 100 %. The NMR spectrum associated with the UiO-66-BA sample shows peaks in the 7.4–8 ppm region, indicating the presence of benzoate ions (Fig. 6). The BDC-to-BA ratio was estimated as 1:0.35. Notably, even after activation at 250 °C, the UiO-66-BA sample still contained benzoate ions, which is consistent with FTIR data. In contrast, the UiO-66-FA sample contained only trace amounts of benzoate ions, as indicated by the very weak NMR peaks. This confirms that treatment with HCl successfully removed benzoate ions from the UiO-66 structure.

Additionally, intense peaks at around 2.3 ppm and 8.5 ppm were observed in the spectra of UiO-66-BA and UiO-66-FA samples activated at RT (Fig. 6a). These peaks correspond to dimethylamine (2.3 ppm) and formate ions (8.5 ppm), indicating that DMF decomposition occurred during sample preparation in the alkaline medium [13]. The presence of these peaks suggests that successive heating at 60 °C, followed by room temperature evacuation, did not completely remove DMF from the UiO-66 pores. This observation is consistent with the FTIR results.

Comparing the samples activated at 250 °C, the UiO-66-BA sample contained only trace amounts of formate ions, while the UiO-66-FA sample showed a significant peak at 8.5 ppm (Fig. 6b). This suggests that formate ions, formed during HCl treatment, replaced benzoate



Fig. 5. Selected regions of IR spectra were measured for (1) UiO-66-BA evac at RT, (2) UiO-66-BA evac at 250 C, (3) UiO-66-BA rehydrated and evacuated at RT; (4) UiO-66-FA evac at RT, (5) UiO-66-FA evac at 250 C, (6) UiO-66-FA rehydrated and evacuated at RT. Green diamonds highlight bands attributed to BA vibrations. Dashed arrows highlight bands corresponding to DMF vibrations.



Fig. 6. (a) Dissolution/<sup>1</sup>H NMR spectra obtained on the UiO-66-BA sample evacuated at RT (1) or at 250  $^{\circ}$ C (2), UiO-66-FA sample evacuated at RT (3) or at 250  $^{\circ}$ C (4). Part (b) shows the aromatic region from 7.0 to 9.0 ppm of the UiO-66-BA (2) and UiO-66-FA (4) samples evacuated at 250  $^{\circ}$ C.

residues in the defect pores. However, the amount of formate ions was about 20 times lower than that of the benzoate ions, possibly due to the removal of formate during activation at 250  $^{\circ}$ C.

It has been reported that the thermal stability of UiO-66-type MOFs strongly correlates with defect concentration [8,30]. Additionally, given our observation of the reversibility of the hydration-dehydration process in the FTIR spectra, we decided to check the stability of the structure using XRD. We slowly heated the UiO-66-BA and UiO-66-FA samples in a dynamic vacuum and activated them at 250 °C for 15 min, replicating the FTIR experiment. After cooling down and refilling the activation vessel with N2, the samples were exposed to air and rehydrated. Fig. 7a shows the XRD profiles of the UiO-66-BA sample before and after activation (for UiO-66-FA, see SI Fig. S7b). This treatment did not alter the crystal structure of the sample. We observed a slight decrease in lattice constants, which we attribute to the evacuation of guest molecules from the pores. The same trend was observed for the UiO-66-FA sample. Thus, we concluded that the activation of synthesized samples did not lead to structural collapse. However, it should be noted that treating these samples in a dynamic vacuum at 250  $^\circ C$  for 6 h or at 275  $^\circ C$  for half an hour resulted in amorphization.

Next, we examined the local coordination of  $Zr^{4+}$  via X-ray absorption spectroscopy. Fig. 7b compares the XANES spectra for assynthesized, activated, and rehydrated UiO-66-BA. It is evident that the main peak position shifts to higher energies after activation. The second spectral feature at 18,080 eV also shifts to higher energy. The intensities of these peaks visibly decrease. Rehydroxylation leads to an inverse effect: the intensities of the main and second peaks slightly

increase, and peak positions shift to lower energies, although they do not return to their initial values. Our results for the intact and activated samples show a trend similar to previous studies on UiO-66 activation [6]. However, we did not observe complete reversibility of dehydration in the XANES spectra. In the present study, the XANES spectral shape does not fully return to its initial state, as seen in the as-synthesized sample. This trend was also observed in the FTIR spectra, where rehydration after the first activation did not result in the complete restoration of the spectral shape and peak positions. We observed shifts in some UiO-66 bands and the disappearance of bands attributed to DMF.

Summarizing all the experimental data from FTIR and XANES spectroscopy and XRD analysis, we conclude that after synthesis, UiO-66 contains DMF molecules inside the pores, which affect the framework behavior (increasing lattice constants, changing vibrations of functional groups, and altering  $Zr^{4+}$  coordination). Activation at 250 °C results in the evacuation of guest molecules, changing the coordination of  $Zr^{4+}$  in the UiO-66 framework. Rehydration results in the adsorption of water molecules, and this process is completely reversible. While short evacuation at 250 °C does not cause structural collapse according to XRD, prolonged treatment can lead to amorphization. We suggest that this correlates with defect concentration, highlighting that the activation procedure is crucial but must be carried out with caution.

# 3.4.2. CO adsorption

Carbon monoxide (CO) is the most commonly used probe molecule for identifying acidic sites in solids, especially in MOFs [17,26,31,32]. We conducted experiments involving low-temperature adsorption of CO on the UiO-66-BA and UiO-66-FA samples evacuated at room temperature (RT) or at 250 °C. This allowed us to compare the active sites in both hydroxylated and dehydroxylated samples.

Fig. 8a represents the spectra of CO adsorbed on UiO-66-BA and UiO-66-FA samples evacuated at RT (see also Fig. S8 in SI). Note that evacuation at RT does not lead to dehydroxylation, and the Zr<sub>6</sub>O<sub>4</sub>(OH)<sub>4</sub> cluster contains  $\mu_3$ -OH groups with Brønsted acidic properties that can form an H-bond with CO. A strong band at 2152  $\text{cm}^{-1}$  dominates in the carbonyl region. This band is assigned to the stretching vibration of the CO molecule H-bonded to the  $\mu_3$ -OH group via the carbon atom:  $\mu_3$ - $OH^{--}C \equiv O$  [17,19,26]. In addition, weak bands at 2131, 2136, and  $2124 \text{ cm}^{-1}$  were observed. The bands at 2131 and 2136 cm $^{-1}$  arise from physically adsorbed CO [17,33]. The weak band at 2124 cm<sup>-1</sup> was attributed to the CO bonded to the  $\mu_3$ -OH group via its oxygen atom:  $\mu_3$ -OH  $\cdots$  O  $\equiv$  C [19,33,34]. The intensity of this band is much lower than for the  $\mu_3$ -OH<sup>...</sup>C  $\equiv$  O adducts, indicating the preferred H-bonding of carbon monoxide through its carbon atom. Adsorption of CO was also accompanied by a shift in the position of the band attributed to vibrations of the  $\mu_3$ -OH group (Fig. 8b) due to the formation of the  $\mu_3$ -OH<sup>...</sup>C  $\equiv$  O adducts. In summary, after evacuation at RT, both samples contained only one type of active site for CO adsorption -  $\mu_3$ -OH groups.



Fig. 7. (a) XRD patterns of as-synthesized UiO-66-BA sample (beige) and the same sample after activation at 250 °C and rehydration (green). (b) In situ X-ray absorption spectra were collected as synthesized (beige), activated (gray), and hydrated (green) UiO-66-BA.



Fig. 8. FTIR spectra registered after low-temperature adsorption of CO (1 mbar equilibrium pressure) on UiO-66-BA and UiO-66-FA samples evacuated at room temperature (a, b) and at 250 °C (c). Part (b) illustrates the difference between the spectrum of the corresponding sample evacuated at room temperature before and after exposure to CO gas at a pressure of 1 mbar at liquid nitrogen temperature.

Evacuation of the samples at 250 °C led to a strong decrease in intensity of the bands around 3674  $\text{cm}^{-1}$  characterizing  $\mu_3$ -OH groups. Hence, the pretreatment resulted in almost full conversion of the  $Zr_6O_4(OH)_4$  cluster to  $Zr_6O_6$ . (Fig. S6). Fig. 8c illustrates the spectra of CO adsorbed on UiO-66-BA and UiO-66-FA samples evacuated at 250 °C. In this case, the most intense bands were those of physically adsorbed CO (2131 and 2136 cm<sup>-1</sup>). Low-intensity bands at 2152 cm<sup>-1</sup> were observed in the spectra of both samples and were attributed to CO adsorbed on residuals of  $\mu_3$ -OH groups. Notably, the UiO-66-BA sample contained a higher amount of these groups, as indicated by the higher intensity of the band at 2152  $\text{cm}^{-1}$ . In contrast, the UiO-66-FA sample contained only small residuals of  $\mu_3$ -OH groups. Consequently, the shift in the position of the band at  $3674 \text{ cm}^{-1}$ , attributed to vibrations of the  $\mu_3$ -OH group due to the formation of  $\mu_3$ -OH...C  $\equiv$  O adducts, was traceable only in the UiO-66-BA sample (Fig. S9 c). No pronounced peaks were observed in the  $3500-3700 \text{ cm}^{-1}$  region in the spectra of the UiO-66-FA sample, likely because they were at the noise level (Fig. S9 d). Thus, the dehydration process was facilitated by removing benzoate residuals from defective pores. It has been reported that larger pores restrain small molecules more weakly, resulting in easier evacuation of physisorbed water [7]. We hypothesize that the removal of benzoate residuals enlarged the defect pores and decreased transport restraints, leading to faster dehydration.

The spectrum of CO adsorbed on the UiO-66-FA sample revealed a new band at 2173 cm<sup>-1</sup>, while the spectrum of the UiO-66-BA sample did not exhibit such a band. The 2173 cm<sup>-1</sup> band was attributed to vibrations of CO molecules bonded to coordinatively unsaturated  $Zr^{4+}$ sites originating from defective pores and formed after removal of the formate ions [17,33]. The absence of such bands in the spectrum of the UiO-66-BA sample indicates that these zirconium sites are blocked by the benzoate modulator and are not able to interact with CO molecules. In contrast, the presence of this band in the UiO-66-FA sample activated at 250 °C demonstrates that post-synthetic modification resulted in the removal of benzoate ions, revealing zirconium sites in defect pores for further interactions (Table 2).

In conclusion, we can infer that the activated UiO-66-FA sample contained accessible  $\mathrm{Zr}^{4+}$  sites, while in the UiO-66-BA sample, zirconium ions in defective pores are saturated with the BA modulator.

#### 3.4.3. CD<sub>3</sub>CN adsorption

Acetonitrile serves as a prevalent probe molecule in FTIR spectroscopy due to its distinctive properties. Possessing base properties, acetonitrile readily interacts with acidic sites, making it an ideal candidate for investigating surface properties. In comparison to carbon monoxide, acetonitrile exhibits a more pronounced basic character. In

#### Table 2

Positions of bands assigned to respective active sites, with values provided in  $cm^{-1}$ . ND indicates when a band was not detected. (vw) denotes a very weak band, and (w) designates a weak band.

| Active site    | μ3-ОН    |                    | Zr <sup>4+</sup> in regular pores |                    | Zr <sup>4+</sup> in defect pore |                    |
|----------------|----------|--------------------|-----------------------------------|--------------------|---------------------------------|--------------------|
| Probe molecule | CO       | CD <sub>3</sub> CN | CO                                | CD <sub>3</sub> CN | CO                              | CD <sub>3</sub> CN |
| Sample         |          |                    |                                   |                    |                                 |                    |
| UiO-66-BA (RT) | 2152     | 2277               | ND                                | ND                 | ND                              | ND                 |
| UiO-66-FA (RT) | 2152     | 2277               | ND                                | ND                 | ND                              | 2303               |
| UiO-66-BA      | 2152 (w) | 2277 (w)           | ND                                | 2300               | ND                              | ND                 |
| (250 °C)       |          |                    |                                   |                    |                                 |                    |
| UiO-66-FA      | 2152     | 2277               | ND                                | 2300               | 2173                            | 2303               |
| (250 °C)       | (vw)     | (vw)               |                                   |                    |                                 |                    |

our experiments, we opted for using deuterated acetonitrile ( $CD_3CN$ ) instead of regular acetonitrile ( $CH_3CN$ ) because of the absence of split of the C-N modes due to Fermi-resonance [26].

Fig. 9a depicts the spectra of CD<sub>3</sub>CN adsorbed on samples activated at room temperature. In both spectra, prominent bands at 2262 and 2277 cm<sup>-1</sup> are observed. These bands are attributed to the C-N stretching modes of physically adsorbed acetonitrile (2262 cm<sup>-1</sup>), and CD<sub>3</sub>CN adsorbed on  $\mu_3$ -OH-groups of Zr<sub>6</sub>O<sub>4</sub>(OH)<sub>4</sub> clusters (2277 cm<sup>-1</sup>) (Table 2). The interaction between  $\mu_3$ -OH-groups and CD<sub>3</sub>CN is also evident from the redshift in the positions of  $\mu_3$ -OH-group bands by 228 cm<sup>-1</sup> induced by the exposure to CD<sub>3</sub>CN (Fig. 9b). This shift, greater than that observed for CO adsorption, underscores the stronger basic properties of acetonitrile.

Additionally, a band at 2303 cm<sup>-1</sup> emerges in the spectrum of UiO-66-FA (Fig. 9a), indicating the interaction of CD<sub>3</sub>CN with  $Zr^{4+}$  ions. The same band is practically absent from the spectrum of UiO-66-BA. Therefore, we infer that these adsorption sites are bare  $Zr^{4+}$  ions in defect pores formed as a result of the removal of modulator anion. Normally, such sites should be blocked by adsorbed water. However, acetonitrile has a relatively high proton affinity (779.2 kJ mol<sup>-1</sup>), higher than that of H<sub>2</sub>O (691 kJ mol<sup>-1</sup>) [19,35], and can replace pre-adsorbed water.

Upon activation at 250 °C, the dehydration process and loss of  $\mu_3$ -OH-groups occur, leading to only faint peaks at 2277 cm<sup>-1</sup> (Fig. 9c). Notably, bands in the 2300-2305 cm<sup>-1</sup> region become visible in the spectra of both samples. It is important to highlight that the peak position for UiO-66-FA has slightly shifted, becoming broader and more intense compared to UiO-66-BA. Therefore, it seems that the acetonitrile peak at ca. 2300 cm<sup>-1</sup> on the UiO-66-FA sample is a composite and



**Fig. 9.** FTIR spectra registered after adsorption of  $CD_3CN$  (3 mbar equilibrium pressure) on UiO-66-BA and UiO-66-FA samples evacuated at room temperature (a, b) and at 250 °C (c). Part (b) illustrates the difference between the spectrum of the corresponding sample evacuated at room temperature before and after exposure to  $CD_3CN$  vapor at a pressure of 3 mbar.

consists of two components (Fig. S12). The higher-frequency component corresponds to adsorption on bare  $Zr^{4+}$  sites in the defect pores and is at least partly detected with the sample evacuated at room temperature. These sites also are able to form carbonyl complexes (see Table 2). In addition, "hidden"  $Zr^{4+}$  sites are detected with both samples at 2300 cm<sup>-1</sup>. These sites are located in the regular pores and are not able to interact with the weak base CO. However, they form complexes with CD<sub>3</sub>CN. Note that a similar situation was previously found with a non-defect UiO-66 sample [19].

The interaction of acetonitrile with  $Zr^{4+}$  ions in regular pores is a rather interesting phenomenon. We measured the XANES spectra of Zr to investigate this interaction, using XANES as an element-selective method. Fig. 10 shows the pairwise comparison of spectra measured at different stages of the experiment.

The first pair shows almost no changes in spectral features after the hydroxylated sample was purged with acetonitrile vapor (Fig. 10a). The second pair clearly depicts shifts in peak intensity and energy position after the activated sample was purged with water vapor (Fig. 10b). The third pair shows changes in the spectral features after the activated sample was purged with acetonitrile (Fig. 10c). It is noteworthy that the changes in the spectra of activated UiO-66-BA after acetonitrile adsorption are similar to those observed after water adsorption. We propose that this indicates acetonitrile molecules change Zr<sup>4+</sup> coordination in a similar way as water molecules do during the rehydration

process, although these changes are less pronounced. Thus, we hypothesize that acetonitrile molecules are adsorbed at the positions of  $\mu_3$ -OH groups, which become vacant after the dehydration process.

# 4. Discussion

Based on the results described above, we propose the following process description. In the first step, we synthesized UiO-66 using benzoic acid as a modulator, resulting in a MOF with defects where zirconium ions in defective pores were coordinated to benzoate anions. These anions are strongly bonded to zirconium and cannot be removed by heating in a dynamic vacuum, thus blocking the zirconium sites in the defect pores and making them coordinatively saturated and inaccessible.

However, the UiO-66-BA sample contains some active sites. Before dehydration, it can provide  $\mu_3$ -OH-groups, which coordinate CO and CD<sub>3</sub>CN probe molecules. After dehydration and the consequent removal of  $\mu_3$ -OH-groups, CO molecules can only be physisorbed. In contrast, CD<sub>3</sub>CN, being a strong base, can interact with Zr sites of the dehydrated UiO-66-BA sample. Given that the Zr-sites in the defective pores of the UiO-66-BA sample are blocked, we conclude that CD<sub>3</sub>CN molecules are adsorbed on Zr-sites in the regular pores, which reduced their coordination number from 8 to 7 after dehydration.

Treating the UiO-66-BA sample with HCl and DMF removed the benzoate residues, resulting in the UiO-66-FA sample. This sample



**Fig. 10.** In situ X-ray absorption spectra collected for rehydrated UiO-66-BA sample before and after  $CH_3CN$  adsorption (a); UiO-66-BA sample before and after evacuation at 250 °C (b); activated at 250 °C UiO-66-BA sample before and after  $CH_3CN$  adsorption (c); in situ X-ray absorption spectra collected at different stages of the experiment (d).

contained formate ions formed through the hydrolysis of DMF in an acidic medium, as shown in the reaction scheme (Fig. 11a) [36]. Formate ions are much less bulky than benzoate ions, increasing accessibility to defective pores. Moreover, during the activation procedure, these residues could be partially decomposed, producing additional bare  $Zr^{4+}$  sites that slightly differ in acidity from the other Zr-sites and can coordinate even weak bases as CO. Thus, this modification opened the Zr-sites in the defective pores.

This results in some alterations in the interaction with probe molecules. Before dehydration, CO was adsorbed only on  $\mu_3$ -OH-groups, similar to the UiO-66-BA sample. However, after dehydration, the Zrsites in the defect pores of the UiO-66-FA sample were able to adsorb CO probe molecules. A similar trend was previously observed for UiO-66, which was synthesized using HCl as a modulator [32]. In this case, the Zr sites in the defective pores became accessible only after thermal treatment.

When a strong base, such as  $CD_3CN$ , was used, we observed that some Zr sites in the defective pores of the UiO-66-FA sample were accessible even in a rehydrated state. For the dehydrated UiO-66-FA sample, we observed adsorption of acetonitrile on Zr sites similar to the UiO-66-BA case. The slightly broadened and shifted band allowed us to conclude that two types of  $Zr^{4+}$  sites can interact with  $CD_3CN$  in the dehydrated UiO-66-FA sample: Zr in defect pores and "hidden" Zr in regular pores.

# 5. Conclusions

In this research, we focused on identifying and assessing the accessibility of active sites within the UiO-66 framework. This is particularly challenging because the UiO-66 framework can undergo reversible changes during dehydration, leading to the reversible disappearance of  $\mu_3$ -OH groups with Brønsted acidic properties and a reduction in the Zr coordination number from 8 to 7. Additionally, UiO-66 can contain defective pores where  $Z^{4+}$  ions, considered active sites with Lewis acidic properties, have accessibility that is highly dependent on synthesis conditions and post-synthetic treatment.

To introduce defects into the structure, we used benzoic acid as a modulator. Benzoic acid is a preferred modulator because it is solid at room temperature, non-corrosive, and does not cause DMF decomposition. Moreover, it forms strong bonds with  $Zr^{4+}$ , resulting in a high defect concentration in the material. However, these strong bonds also block Zr in defective pores. We used *in-situ* FTIR analysis with CO and acetonitrile probe molecules to trace the accessibility of  $Zr^{4+}$  ions. After synthesis, benzoate anions are coordinated to  $Zr^{4+}$  ions, thus blocking them for adsorption. Standard washing and heating procedures did not remove these benzoate residues from the defective pores of UiO-66. Although the samples with benzoate ions in defective pores could be dehydrated with the loss of  $\mu_3$ -OH groups, they retained the benzoate ions.

We observed that the dehydrated sample with benzoate residues was still capable of adsorbing acetonitrile molecules. Since the  $Zr^{4+}$  ions in defective pores were blocked, we inferred that the acetonitrile was adsorbed on  $Zr^{4+}$  ions in regular pores, where the coordination number had been reduced. Using *in-situ* XANES analysis, we monitored changes in the coordination sphere of zirconium in regular pores upon acetonitrile adsorption, finding that these changes were similar to those observed after water adsorption. This led us to conclude that acetonitrile occupies the vacant positions left by  $\mu_3$ -OH groups. It is important to note that these Zr sites are present in any activated UiO-66, regardless of whether a linker or modulator is coordinated to the zirconium ions. Therefore, this hidden acidity is not associated with structural defects in UiO-66. In contrast, true acidity only occurs in locations where benzoates have been removed, reducing the number of ligands at the node compared to the ideal structure.

Next, we treated the samples with HCl and DMF to remove benzoate residues from the defective pores. DMF hydrolysis resulted in the



**Fig. 11.** (a) DMF hydrolysis in an acidic medium. (b) UiO-66 modification: HCl and DMF treatment with heating replaces benzoate residues in defect pores with formate. Gray spheres represent carbon, dusty-pink spheres represent oxygen, and dashed lines represent carboxylic bonds. Zirconium is coordinated with either benzoate or formate residues.

formation of formic acid, which coordinated with Zr sites in the defect pores instead of benzoate residues. This substitution was traced using IR and Raman spectroscopy. Formate ions are much less bulky and can be partially decomposed during heating. In-situ FTIR analysis with CO and acetonitrile probe molecules showed that Zr ions in defective pores were now accessible and served as sites for the adsorption of probe molecules.

In conclusion, active sites in UiO-66 are highly desirable for imparting acidic properties to the framework. However, this MOF is a complex system whose active sites can vary depending on the synthesis and post-synthetic treatment. Site-specific methods such as *in-situ* IR analysis with probe molecules and XANES are essential for understanding these changes. Comprehensive analysis is crucial to accurately predicting and identifying active sites in UiO-66. Moreover, we precisely traced the process of UiO-66 modification, which allowed the introduction of Zr sites with increased Lewis acidity. These accessible sites can be used for attaching functional groups with specific properties to the MOF.

#### CRediT authorship contribution statement

Vera V. Butova: Writing – review & editing, Writing – original draft, Investigation, Formal analysis. Videlina R. Zdravkova: Investigation, Formal analysis. Olga A. Burachevskaia: Investigation, Formal analysis. Ivan E. Gorban: Investigation, Data curation. Mikhail A. Soldatov: Writing – review & editing, Validation, Methodology, Formal analysis. Konstantin I. Hadjiivanov: Writing – review & editing, Supervision, Funding acquisition, Data curation, Conceptualization.

#### Declaration of competing interest

The authors declare the following financial interests/personal relationships which may be considered as potential competing interests:

Vera Butova, Videlina Zdravkova, and Konstantin Hadjiivanov report that the Bulgarian Science Fund provided financial support. Mikhail Soldatov reports that the Southern Federal University provided financial support. Olga Burachevskaia reports that the Ministry of Science and Higher Education of the Russian Federation provided financial support.

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# Appendix A. Supplementary data

Supplementary data to this article can be found online at https://doi.org/10.1016/j.micromeso.2024.113372.

# Data availability

Data will be made available on request.

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